



# Leakage Current Reduction in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky Barrier Diodes by CF<sub>4</sub> Plasma Treatment

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**Abstract**—This letter reports on the suppression of reverse leakage current ( $I_r$ ) in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky barrier diodes (SBDs) through Schottky barrier modification by a low power CF<sub>4</sub>-plasma treatment prior to Schottky metal deposition. Revealed by an x-ray photoelectron spectroscopy (XPS) analysis, the fluorine-plasma treatment brought an incorporation of fluorine ions and depletion of silicon donors in the near surface region of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>, and thus raised its surface potential by around 0.14 eV. Furthermore, insulating GaF<sub>x</sub> was likely created at the Schottky interface. Attributed to the fluorine-plasma-modified Schottky barrier, a reduced  $I_r$  by around four orders of magnitude and enhanced blocking voltage ( $V_{block}$ ) from 150 V to 470 V at  $I_r = 100 \mu\text{A}/\text{cm}^2$  have been achieved without degrading the forward characteristics. Different from the untreated device whose  $I_r$  was purely governed by the thermionic field emission (TFE), the fluorine-plasma-treated SBD showed a greatly suppressed TFE-current until a space-charge limited current (SCLC) started to dominate at around  $-500 \text{ V}$ .

**Index Terms**— $\beta$ -Ga<sub>2</sub>O<sub>3</sub>, plasma treatment, Schottky barrier diodes, reverse leakage current, space-charge limited current.

## I. INTRODUCTION

IN RECENT years,  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> semiconductor has attracted great research interest due to its superior material properties. The ultra-wide bandgap of  $\sim 4.8 \text{ eV}$ , which corresponds to a wavelength of about 260 nm, makes  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> an intrinsically suitable candidate towards high performance solar-blind deep-ultraviolet (DUV) detectors [1]–[3]. Featuring a relatively large atom density and good radiation tolerance,  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> is

also promising for X-ray and high-energy particle detection applications [4]–[6]. Moreover,  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> shows a great potential for next-generation high-voltage power electronics, since its critical electric field is up to 8 MV/cm and the projected Baliga's figure of merit (BFoM) exceeds 3 GW/cm<sup>2</sup> [7], [8].

As a fundamentally important device, Schottky barrier diodes (SBDs) based on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> have been extensively developed for all aforementioned applications. Remarkably, over 1 kV breakdown voltage ( $V_B$ ) has been achieved in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> power SBDs by many researchers through careful electric field management [9]–[12]. However, a relatively high reverse leakage current ( $I_r$ ) in the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs, which mainly results from an emission of majority carriers over a relatively low potential barrier (typically  $\sim 1 \text{ eV}$ ), remains a critical challenge for both power switching and sensing applications, since it will lead to a large power consumption, high noise level, low efficiency and reliability issues. Therefore, leakage current reduction and Schottky barrier modulation are essential for developing better-performance  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky devices [13].

A previous study showed that exposure of the Pt/ $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky interface to a hydrofluoric acid could increase its Schottky barrier height and decrease the  $I_r$ , in which a diffusion of fluorine (F) atoms into  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> was claimed but lacked direct confirmation [14]. On the other hand, F-based plasma treatments have been successfully adopted in GaN-based SBDs [15]–[17] and high electron mobility transistors (HEMTs) [18]–[20] for effective Schottky barrier modulation. However, performance degradation was recently observed in CF<sub>4</sub>-plasma-treated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs [21], [22], in terms of decreased forward current and increased ideality factor and  $I_r$ . In these studies, surface damage was likely generated due to the non-optimized treatment conditions, and it could be partially recovered by a subsequent annealing process.

In this work, through a low power CF<sub>4</sub>-plasma surface treatment, we successfully achieved Schottky barrier modification in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs and reduced the  $I_r$  by around four orders of magnitude, while maintaining good forward characteristics. An x-ray photoelectron spectroscopy (XPS) study confirmed the incorporation of F ions and formation of insulating GaF<sub>x</sub> at the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky contact after treatment.

## II. DEVICE STRUCTURE AND FABRICATION

The sample used in this study consisted of an 8- $\mu\text{m}$  thick Si doped  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> drift layer grown on a conductive (001)  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> substrate. The non-linear  $I/C^2$ -V plot in Fig. 1(a) indicated a non-uniform doping in the drift layer. The net

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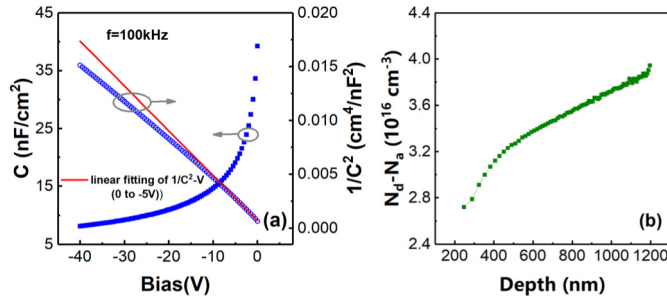


Fig. 1. (a) The measured  $C$ - $V$  and  $1/C^2$ - $V$  curves of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBD without F-plasma treatment. (b) The extracted net doping concentration ( $N_D - N_A$ ) from  $C$ - $V$  measurement.

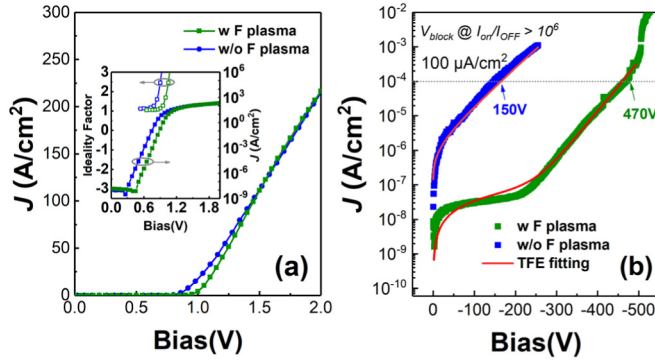


Fig. 2. Comparison of (a) forward and (b) reverse  $I$ - $V$  characteristics for the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs with and without F-plasma treatment.

doping concentration ( $N_D - N_A$ ) was determined to be  $2.8 \sim 4 \times 10^{16} \text{ cm}^{-3}$ , as shown in Fig. 1(b). The fabrication process of the SBDs began with sample cleaning using acetone and isopropanol, followed by a deionized (DI) water rinse. After cleaning, the Ohmic cathode electrode (Ti/Al/Au) was deposited on backside of the sample by blanket e-beam evaporation. Then the sample was loaded into a reactive ion etch (RIE) system and a CF<sub>4</sub>-plasma treatment was performed on the front side of the sample for 1 minute with a low RF power of 50 W. The chamber pressure and CF<sub>4</sub> gas flow rate were 10 mTorr and 25 sccm, respectively. Finally, circular-shaped anode electrodes with a diameter of 100  $\mu\text{m}$  were formed by evaporating the Schottky metal (Ni/Au) onto the sample surface. To make comparison, a control sample without going through the F-plasma treatment process was co-prepared. Prior to the Schottky metal deposition, XPS measurements were performed on both samples to reveal how the F plasma affected the surface condition and chemical composition of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>.

### III. RESULTS AND DISCUSSION

Fig. 2(a) plots the forward current-voltage ( $I$ - $V$ ) curves of the fabricated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs with and without F-plasma treatment. The two devices showed quite comparable on-state performances with close-to-unity ideality factors and the forward current densities both exceeding 0.2 kA/cm<sup>2</sup> at a bias of +2 V. A similar specific on-resistance ( $R_{on,sp}$ ) of  $\sim 4.6 \text{ m}\Omega \cdot \text{cm}^2$  was obtained. The turn-on voltages ( $V_{on}$ ), extracted at a current density of 1 A/cm<sup>2</sup>, were 0.95 V and 0.82 V for the F-plasma-treated and untreated SBDs, respectively.

Fig. 2(b) compares the reverse  $I$ - $V$  characteristics of the fabricated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs with and without F-plasma

TABLE I  
BENCHMARK OF KEY PARAMETERS IN Ga<sub>2</sub>O<sub>3</sub> SBDs

Ref.	$N_D$ [cm <sup>-3</sup> ]	$R_{on,sp}$ [m $\Omega$ ·cm <sup>2</sup> ]	$I_{ON}/I_{OFF}$ @ -200V	$V_{block}$ [V] @ 100 $\mu\text{A}/\text{cm}^2$	E-field manag.
[9]	$3 \times 10^{16}$	2.0	$5 \times 10^6$ $1 \times 10^2$	230 60	FP
[10]	$1 \times 10^{16}$	4.7	$2 \times 10^{10}$	1100	GR+FP
[11]	$1.5 \times 10^{16}$	5.5	$1 \times 10^6$ $4 \times 10^6$	140 290	ET
[12]	$3 \times 10^{16}$	3.0	$4 \times 10^8$	550	ET
[13]	$4 \times 10^{16}$	2.9	$7 \times 10^5$	100	No
[14]	$1.8 \times 10^{16}$	4.2	$9 \times 10^6$	240	No
[14]	$1.8 \times 10^{16}$	5.1	$3 \times 10^{10}$	950	FP
This work	w/o F w F	$3.5 \times 10^{16}$ 4.6	$7 \times 10^5$ $4 \times 10^9$	150 470	No No

\*Field Plate (FP); Guard Ring (GR); Edge Termination (ET).

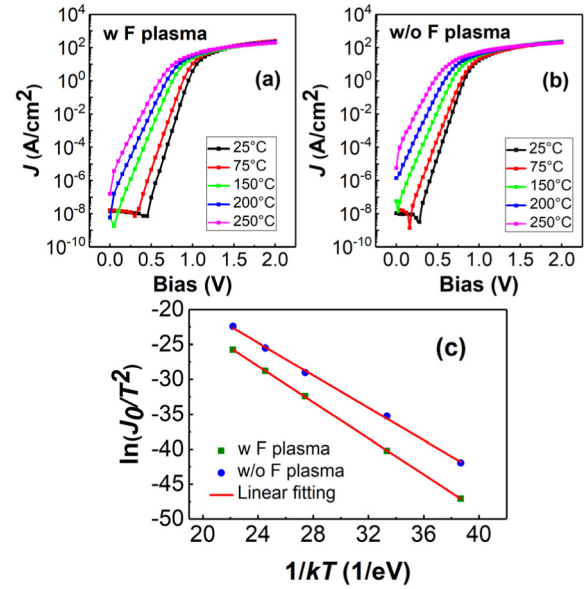


Fig. 3.  $T$ -dependent forward  $I$ - $V$  characteristics of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs (a) with and (b) without F-plasma treatment, and (c) the corresponding Richardson's plots in a temperature range of 25 to 250 °C.

treatment in a semi-log scale. The  $I_r$  in the untreated SBD increased obviously with the increase of the bias voltage and reached to  $\sim 3 \times 10^{-4} \text{ A/cm}^2$  at -200 V. On the other hand, the F-plasma-treated SBD exhibited a  $I_r$  of  $\sim 5 \times 10^{-8} \text{ A/cm}^2$  at -200 V, approximately 4 orders of magnitude lower than the untreated one. Therefore, an On/Off current ratio ( $I_{ON}/I_{OFF}$ ) at -200 V as high as  $4 \times 10^9$  was achieved. Moreover, the blocking voltage ( $V_{block}$ ) of the device, which was defined at an  $I_r$  of 100  $\mu\text{A}/\text{cm}^2$  and corresponded to an  $I_{ON}/I_{OFF} > 10^6$ , was significantly enhanced by the proposed F-plasma treatment from 150 V to 470 V, among the highest in all reported  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs.

Table I benchmarks the device characteristics in our study with the other state-of-the-art  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs. The outstanding reverse blocking capability and well-maintained on-state performance in our device indicated that Schottky barrier modification was achieved by the proposed F-plasma treatment without degrading the quality of the Schottky contact. It should be noted that the chamber pressure in our CF<sub>4</sub>-plasma treatment process was much lower than that in previous reports [21], [22]. The lower chamber pressure led to

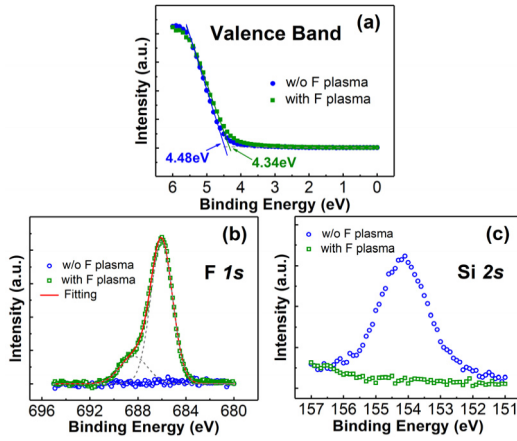


Fig. 4. (a) Valence band spectra, (b) F 1s core-level spectra and (c) Si 2s core-level spectra of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples with and without F-plasma treatment.

fewer ion-substrate collisions during treatment and less plasma damage onto the device [23].

The temperature ( $T$ )-dependent forward  $I$ - $V$  characteristics of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs with and without F-plasma treatment are shown in Fig. 3, alongside with the corresponding Richardson's plots. The F-plasma-treated SBD showed an increased Schottky barrier height (SBH) of 1.31 eV compared to the untreated device of 1.18 eV, in good agreement with the 0.13 V positive shift in  $V_{on}$ . Compared to a theoretical value taking the electron effective mass of  $0.342 m_0$  [24], the effective Richardson's constant ( $A^*$ ) extracted in our devices ( $20 \sim 23 \text{ A/cm}^2\text{K}^2$ ) was relatively small. The effects of a high tunneling current could be a possible reason for the untreated SBD [25], [26], while the low  $A^*$  in the F-plasma-treated device suggested a presence of dielectric interfacial layers [27], probably GaF<sub>x</sub> as discussed later.

Previous reports showed that the F-plasma treatment on GaN-based devices suffered from thermal stability issues, for example, threshold voltage shifting at elevating temperatures [28]. In our study, no performance degradation was observed for the F-plasma treated Ni/ $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs after the high temperature test up to 250 °C. However, higher temperature stability has not been studied yet and further investigation is needed.

As shown in the valence band spectra in Fig. 4 (a), the F-plasma treatment shifted the surface Fermi level in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> toward the valence band by  $\sim 0.14 \text{ eV}$ , which corresponded to an increase in the surface potential and should be responsible for the higher SBH and  $V_{on}$  observed in the F-plasma-treated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBD. Fig. 4 (b) and (c) compare the F 1s and Si 2s core-level spectra, respectively, between the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples with and without F-plasma treatment. Apparently, the F 1s peak located at  $\sim 686.0 \text{ eV}$  with a shoulder on the higher binding energy side, which corresponded to GaF<sub>x</sub> [29], appeared after F-plasma treatment, confirming an incorporation of F ions into the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> and the formation of insulating GaF<sub>x</sub> at the surface [15], [20]. On the other hand, the Si 2s peak vanished in the F-plasma-treated sample, suggesting a removal of silicon (Si) dopant from the near surface region of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>. During treatment, the CF<sub>4</sub> plasma likely reacted with Si atoms to form volatile SiF<sub>4</sub>, which was then flushed out of the RIE chamber. Thus, it was concluded that the proposed F-plasma treatment could simultaneously enlarge the effective SBH and

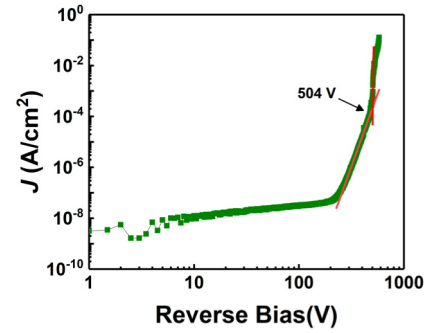


Fig. 5. The  $I_r$  of the F-plasma-treated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs plotted in a full-log-scale.

surface depletion width in a  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBD, mainly due to the strong electronegativity of the incorporated F ions together with a depletion of Si dopants. The removal of Si dopant occurred only in the near surface region, much shallower than a Ni/ $\beta$ -Ga<sub>2</sub>O<sub>3</sub> Schottky depletion depth, which would hardly affect the SBD's forward conduction.

As confirmed by the curve fitting in Fig. 2(b), a pure thermionic field emission (TFE) model well described the  $I_r$  in the untreated SBD with an SBH of 0.79 eV [32]. This value was lower than the SBH extracted from the forward  $I$ - $V$  characteristics (1.18 eV), which could be caused by the injection of electrons into the defect mini-band rather than the conduction band [33] or the inhomogeneity of the Schottky barrier height [34], [35].

On the other hand, the  $I_r$  in the F-plasma-treated SBD showed a much smaller dependency on the bias voltage below  $-240 \text{ V}$  and a hump at around  $-500 \text{ V}$ , suggesting a different leakage mechanism from the untreated device. Such improvement was attributed to not only a higher and wider Schottky barrier but also the influence of the insulating GaF<sub>x</sub>, which effectively blocked the tunneling current and might also passivate the interface states [16]. A parallel leakage path dominated the measured  $I_r$  in the F-plasma-treated SBD at a low reverse bias ( $< 240 \text{ V}$ ), and the  $I_r$  could be fitted well up to a reverse bias of 500 V based on the TFE model including a fixed specific parallel resistance of  $3 \times 10^9 \Omega\text{cm}^2$ . The fitted SBH value for the F-plasma-treated SBD was 1.09 eV, obviously larger than that of the untreated device (0.79 eV), confirming a successful Schottky barrier modification by the proposed F-plasma treatment.

Fig. 5 plots the  $I_r$  of the F-plasma-treated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBD in a full-log scale. It was found that the  $I_r$  showed a sudden increase at around  $-500 \text{ V}$ , which could be modeled by a space-charge limited current (SCLC) conduction near the traps-filled-limit voltage ( $V_{TFL}$ ) [30], [31], [33]. Under a high-level reverse bias, the electrons that injected into the depletion region could be partly captured by traps until the  $V_{TFL}$  was reached, at which point a soft breakdown usually occurred.

#### IV. CONCLUSION

A CF<sub>4</sub>-plasma treatment process has been developed for leakage current reduction in  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> SBDs. Through an incorporation of F ions and removal of Si dopants, as well as a formation of insulating GaF<sub>x</sub> at the Schottky interface, Schottky barrier modification was realized. The F-plasma-treated device showed significantly enhanced reverse characteristics including a  $\sim 10^4 \times$  lower  $I_r$  and  $> 3 \times$  higher  $V_{block}$ , while maintaining good on-state performances.



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